

# Improvement of sintering and magnetic properties of nanocrystalline Mn-Zn ferrites

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**Abstract:** The effect of sintering temperature on the formation of Mn-Zn ferrites in a single phase cubic spinel structure has been investigated. Samples of general chemical formula  $Mn_{1-x}Zn_xFe_2O_4$ , ( $0.0 \leq x \leq 0.5$ ) were prepared using standard ceramic technique. The obtained results show that 1300°C for 15h with heating rate 20°C/min are suitable conditions for the preparation of Mn-Zn ferrites. The effect of the double final sintering and  $Zn^{2+}$  ions concentration on the physical properties of Mn-Zn ferrite was also investigated. XRD data confirm the formation of the samples in single phase cubic spinel structure. The crystallite size ( $t$ ) of samples was estimated by Scherrer's formula and found in the range (90-115 nm). Energy dispersion X-ray analysis (EDX) was used for the elemental analysis of the prepared samples. Magnetization of the samples was measured using Faraday's method as a function of temperature at magnetic field 1300 Oe. The Curie temperature of the prepared samples was found to be decreases with increasing  $Zn^{2+}$  ion. The hysteresis parameters such saturation magnetization  $M_s$  and coercivity  $H_c$  of the prepared samples were measured at room using vibrating sample magnetometer (VSM). An improvement in the physical properties of Mn-Zn ferrite was observed due to the preparation conditions compared with the published data.

**Keywords:** Mn-Zn Ferrites, Sintering, EDX, VSM; Curie temperature; Magnetization, Hysteresis Parameters.



## 1. Introduction

Ferrites are ceramic materials of high magnetic properties and they are mainly used as a core for inductors, transformers and recording heads [1-3]. The resistivity of Mn-Zn ferrite is relatively small with high eddy current loss, so many methods of preparation was used to improve its physical properties. Mn-Zn ferrites have been studied by many authors with different preparation methods such as chemical co-precipitation method by E.Veena et.al [4], hydrothermal precipitation processing by Y.Xuan et. al [5] and reverse micelle technique by P. Poddor et.al [6]. Also D. Ravinder et. al studied the dc electrical resistivity of Mn-Zn ferrite as function of temperature [7].

It is well known that, the cation distribution and consequently the physical properties of the ferrites depends on the preparation conditions such as the sintering temperatures and the cooling rate [8]. In our previous work we study the effect of heating rate on

the electrical properties of Mn-Zn ferrites and the obtained results shown that, the heating rate controlled all the electrical characteristics of the samples due to the change in the grain size [9]. The saturation magnetization of the Mn-Zn ferrites depends on the solubility of the cations  $Mn^{2+}$  and  $Zn^{2+}$  ions in the ferrite lattice and the occupying positions of tetrahedral or octahedral sites [10]. So, the preparation method and the preparation conditions are very important to control the magnetic properties of Mn-Zn ferrites.

In present work we study the effect of sintering temperature on the formation of Mn-Zn ferrites in a single phase cubic spinel structure. Moreover, the present work investigates the effect of double sintering and  $Zn^{2+}$  ions concentration ( $x$ ) on the magnetic properties of Mn-Zn ferrites.

## 2. Experimental Techniques

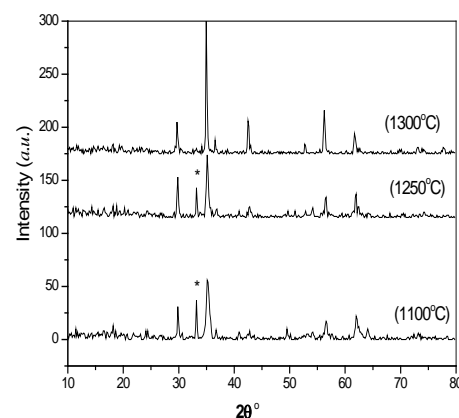
Ferrites with general chemical formula  $Mn_{1-x}Zn_xFe_2O_4$ , ( $0.0 \leq x \leq 0.5$  step 0.1) were prepared

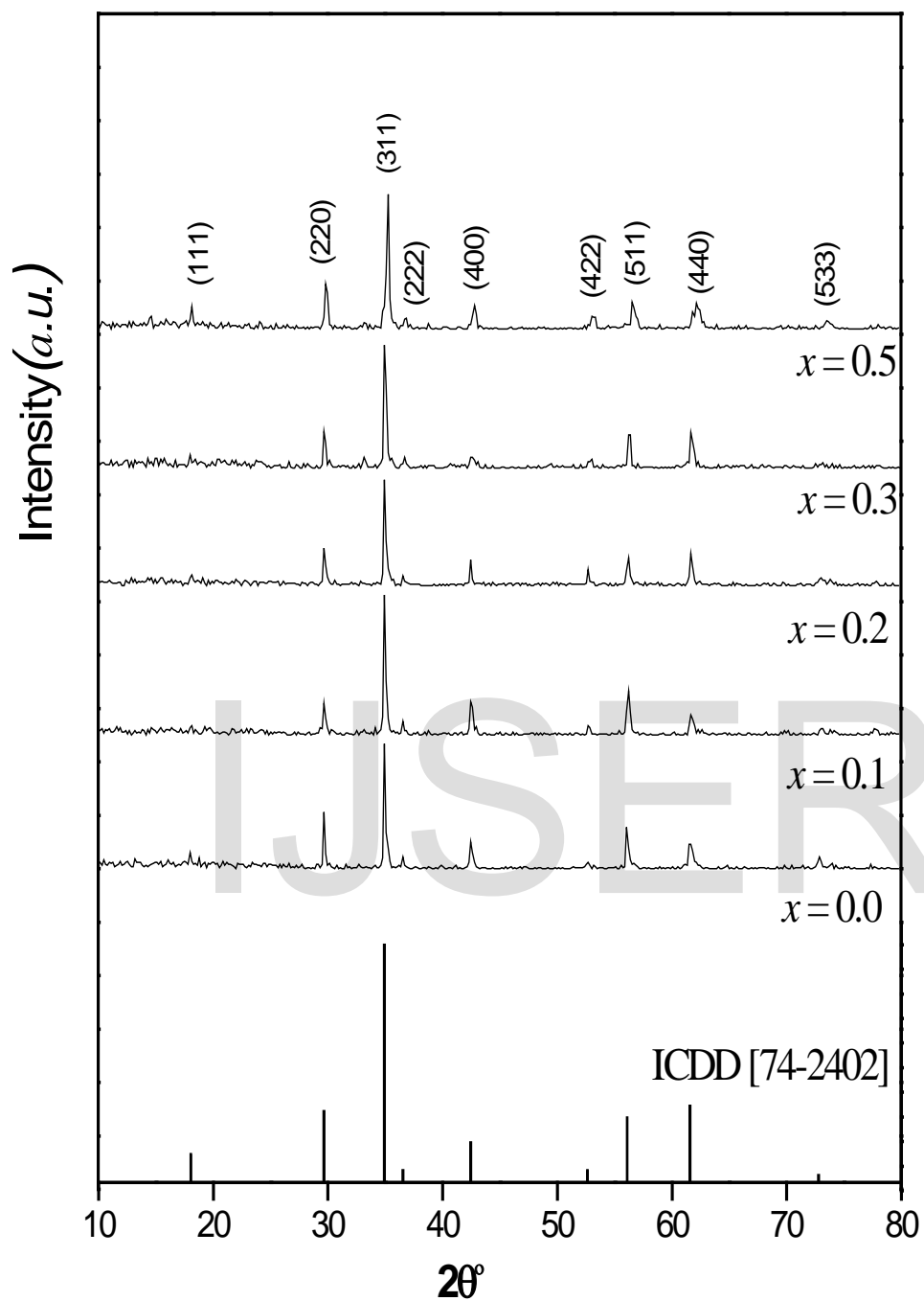
using the standard ceramic technique. High purity oxides in stoichiometric ratio were used and well grinded for 4 h and pre-sintered in air at 900°C for 5h with heating rate 4 °C/min using furnace then slow cooled to room temperature. After that, the samples were ggrinded again for 3 h and the mixture was pressed into pellets using uni-axial press of pressure  $1.9 \times 10^8 \text{ Nm}^{-2}$ . The sample of  $\text{Mn}_{0.9}\text{Zn}_{0.1}\text{Fe}_2\text{O}_4$  was final sintered in air in a closed oven at three different temperatures 1100°C, 1250°C and 1300 °C for 15h to study the effect of sintering temperature on the formation of single phase Mn-Zn ferrites. The final sintering of all the samples was carried out in two steps at 1100 °C for 10h then at 1300 °C for 15h with a heating rate 2°C/min. Crystalline phases in of the different samples were identified using XRD on a Brucker axis D8 diffractometer using  $\text{Cu-K}\alpha$ . The elemental analysis was performed using energy dispersive X-ray (EDX). The magnetization of the investigated samples was measured using Faraday's method. Also, the hysteresis loop parameters were measured using the vibrating sample magnetometer (VSM) of a magnetizing field ranged from 0 to 6000 Oe.

### 3. Results and Discussion

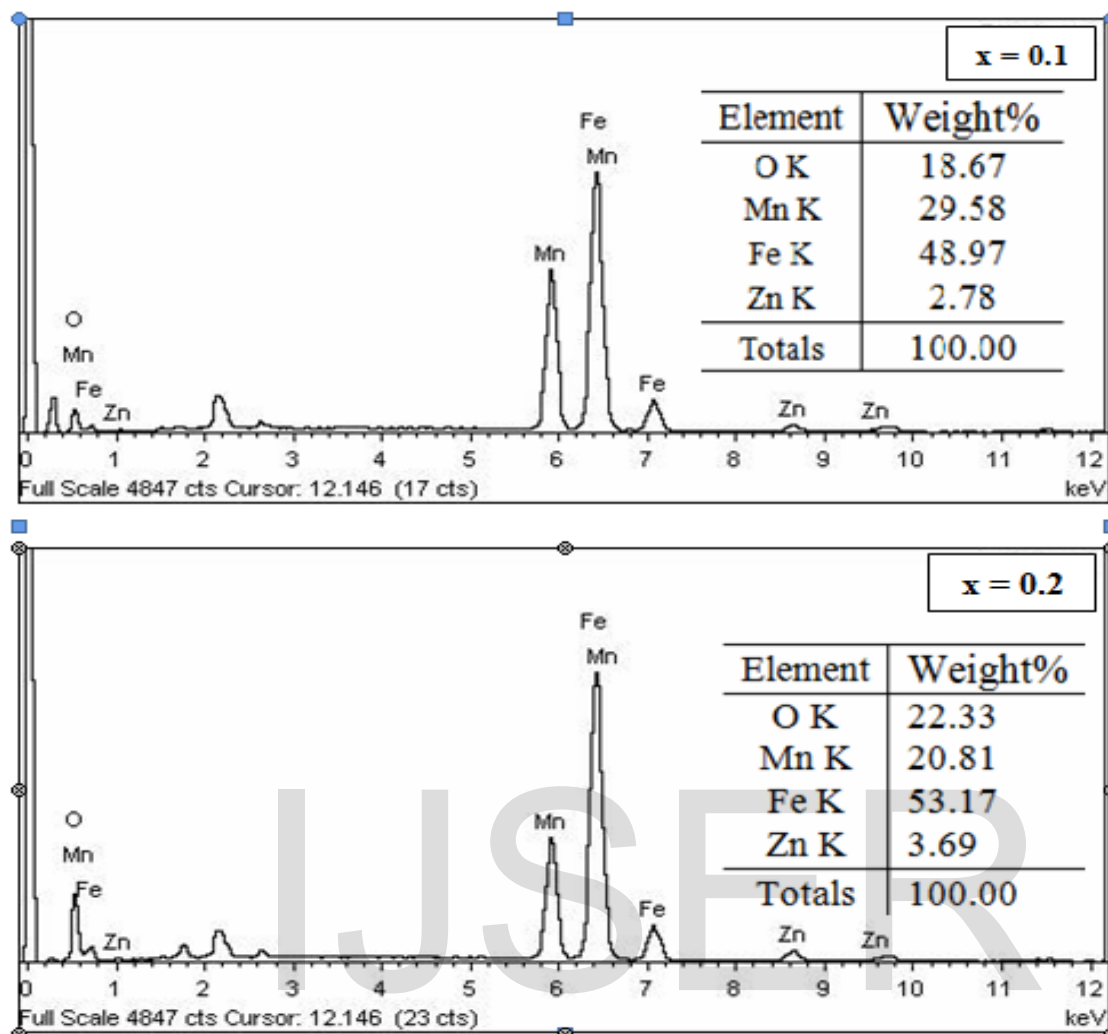
#### 3.1. Structural properties

Crystal structure of all the prepared samples was examined using XRD patterns. The XRD patterns of the prepared samples at different sintering temperatures 1100, 1250 and 1300 °C are shown in Fig.1. Figure 1 shows that, the samples prepared at 1100 and 1250 °C exhibit a peak of  $\text{Fe}_2\text{O}_3$  secondary phase but the prepared sample at 1300 °C has a single phase cubic spinel structure. The obtained results investigate that, the final sintering temperature at 1300 °C for 15h with heating rate 2 °C/min are suitable conditions for the preparation of single Mn-Zn ferrites. Fig. 2 shows the XRD patterns of all the prepared ferrite samples. The figure indicates that the samples are have a single phase cubic spinel structure where the crystallographic planes of the characteristic peaks of cubic spinel structure are perfectly matched with the theoretical data of Franklinite spinel structure (ICDD card No. 74-2402).





**Fig. 2:** X-ray diffraction patterns of all the samples  $\text{Mn}_{1-x}\text{Zn}_x\text{Fe}_2\text{O}_4$



**Fig. 3:** EDX data for the samples of  $x = 0.1$  and  $x = 0.2$ .

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he crystallite size ( $t$ ) of the prepared samples was estimated using Scherrer's equation [11] and the calculated crystallite size ( $t$ ) as a function of Zn content ( $x$ ) shows that, the substitution of  $Mn^{2+}$  ions by  $Zn^{2+}$  ions results in a decrease in the crystallite size from 115 nm to 90 nm. Compositional analysis using energy-dispersive X-ray spectroscopy (EDX) was

these compositions and the obtained results are shown in shown in Fig. 3. The ratios as determined by EDX measurements for  $x = 0.1$  and  $x = 0.2$  are compared with the prepared ratios as shown in Table [1]. Table [1] shows that, the stoichiometric values are very close to the measured values.

**Table 1:** EDX measurements for samples  $x = 0.1$  and  $x = 0.2$ .

Sample	ratio	Prepared ratio	Average EDX ratio
$\text{Mn}_{0.9}\text{Zn}_{0.1}\text{Fe}_2\text{O}_4$	Mn:Fe	0.45	0.5
	Zn:Fe	0.05	0.056
$\text{Mn}_{0.8}\text{Zn}_{0.2}\text{Fe}_2\text{O}_4$	Mn:Fe	0.4	0.39
	Zn:Fe	0.1	0.11

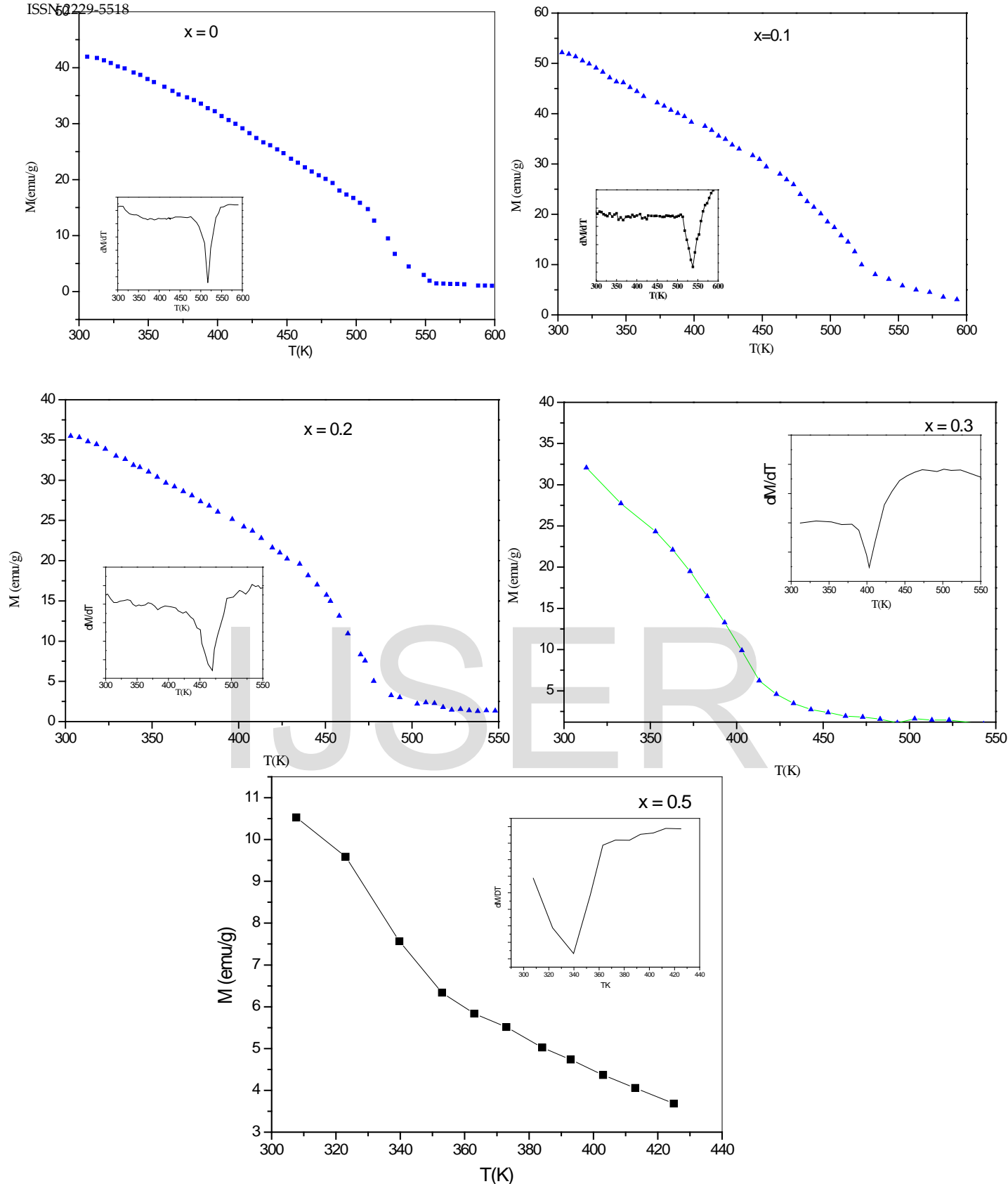
## 3.2. Magnetic properties

### 3.2.1. Magnetization

The magnetization (M) of the prepared samples was measured as a function of temperature and the obtained results are shown in Fig. 4. Also the temperature dependence of the Magnetization for all studied samples at 1300 Oe is shown in Fig. 5. Figures 4, 5 show a normal ferrimagnetic character as the general trend for all the spinel ferrites where the magnetization of the prepared samples decreases gradually with increasing the temperature until the Curie temperature  $T_C$  after which the sample shows a paramagnetic trend [12, 13].

Figure 6 shows the compositional dependence of the magnetization at room temperature. This figure shows that, as Zn-content increases M increases up to  $x = 0.1$  and then decreases. The increase in the magnetization with increasing Zn-content is attributed to  $\text{Zn}^{2+}$  ions of zero magnetic moment should occupy the A-site instead of  $\text{Mn}^{2+}$  ions of magnetic moment

$5.85\mu_B$ , then the magnetization of A-site decreases. According to the Neel's two sublattice collinear model, the net magnetization ( $M = M_B - M_A$ ) increases and the maximum magnetization was obtained at  $x = 0.1$ . But with more increase of Zn-content M decreases, by assuming a combination of parallel and anti-parallel moments, since Zn-ferrite is known to be paramagnetic, it is expected that the B-B interaction will be antiferromagnetic even in the mixed Zn ferrite. The effect of this interaction is usually masked by strong A-B interaction which causes the spins in B-site to be aligned parallel to each other. If, however, the A-B interaction is weakened by introducing non-magnetic  $\text{Zn}^{2+}$  ions into A-site then the tendency toward an antiferromagnetic arrangement in the B-site will be increased [14]. For high  $\text{Zn}^{2+}$  concentration Yafet-Kittel model is obeyed in which the nonlinearity in the B-site is reflected by Y-K angle between the moments in the B-site which is responsible for the deviation from the Neel's collinear model. So for  $x \geq 0.2$ , the magnetization decreases due to the increases in the Y-K angle ( $M = M_B \cos\theta_{YK} - M_A$ ) [15, 16].

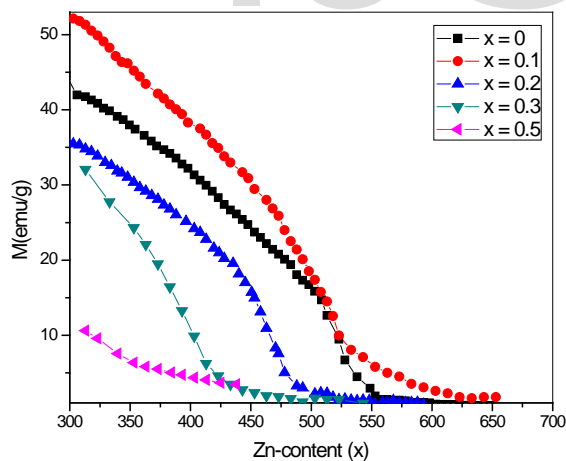


**Fig. 4:** The variation of the magnetization and the onset  $\frac{dM}{dT}$  as a function of absolute temperature  $T$ (K) at a magnetic field 1300Oe for  $Mn_{1-x}Zn_xFe_2O_4$  system; ( $0.0 \leq x \leq 0.5$ ).

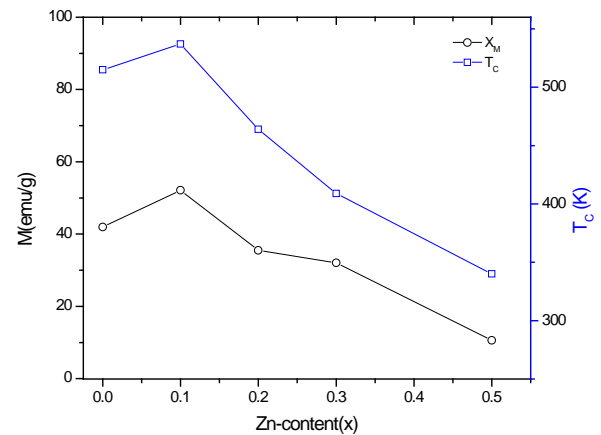
The Curie temperature of the prepared samples was determined by plotting the first derivative of the

magnetization  $\frac{dM}{dT}$  as a function of absolute

temperature, as an inset in Fig.4. The compositional dependence of the Curie temperature ( $T_C$ ) is shown in Fig. 6. The figure shows that, the Curie temperature ( $T_C$ ) decreases with increasing Zn content ( $x$ ). The decrease in the Curie temperature is due to the substitution of  $Mn^{2+}$  ions by non-magnetic  $Zn^{2+}$  ions at tetrahedral A-site. The substitution of nonmagnetic  $Zn^{2+}$  ions reduces the active magnetic linkages per magnetic ion per formula unit; as a result ( $T_C$ ) decreases with increasing Zn-content [17]. In other words the increase in the non-magnetic Zinc ions reduces the magnetic moment of the A- sub-lattice as the  $Zn^{2+}$  ions preferentially occupy the A- sites of the spinel lattice thereby reducing the A-B exchange interaction between A and B sub-lattices, this results in a reduction in the Curie temperature [18].



**Fig. 5:** Magnetization vs the absolute temperature  $T(K)$  for the samples  $Mn_{1-x}Zn_xFe_2O_4$ ; ( $0.0 \leq x \leq 0.5$ ) at 1300 Oe.



**Fig. 6:** variation of the Magnetization ( $M$ ) and Curie temperature vs  $Zn^{2+}$  content ( $x$ ) at room temperature.

### 3.2.2. Vibrating sample magnetometer

The hysteresis parameters of the prepared samples were measured at room temperature and the hysteresis curves are shown in Fig. 7. The figure shows a hysteresis in the samples indicating a ferromagnetic behavior where the magnetization increases with increasing the applied magnetic field and attains its saturation value at higher field. The effect of the substitution by  $Zn^{2+}$  ions on the magnetization  $M_s$ , remanence  $M_r$  and coercivity  $H_c$  of the prepared samples at room temperature and maximum field of 6000 Oe are summarized in Table [2]. The small values of the remanence magnetization confirm the soft character of the samples. The obtained results show that, as Zn content increases the saturation magnetization decreases, this is expected on basis of the substitution of  $Mn^{2+}$  ions by nonmagnetic  $Zn^{2+}$  ions.

The high values of the coercivity  $H_c$  of the prepared samples means that the samples have high resistance to

be demagnetized which is useful in the magnetic applications.

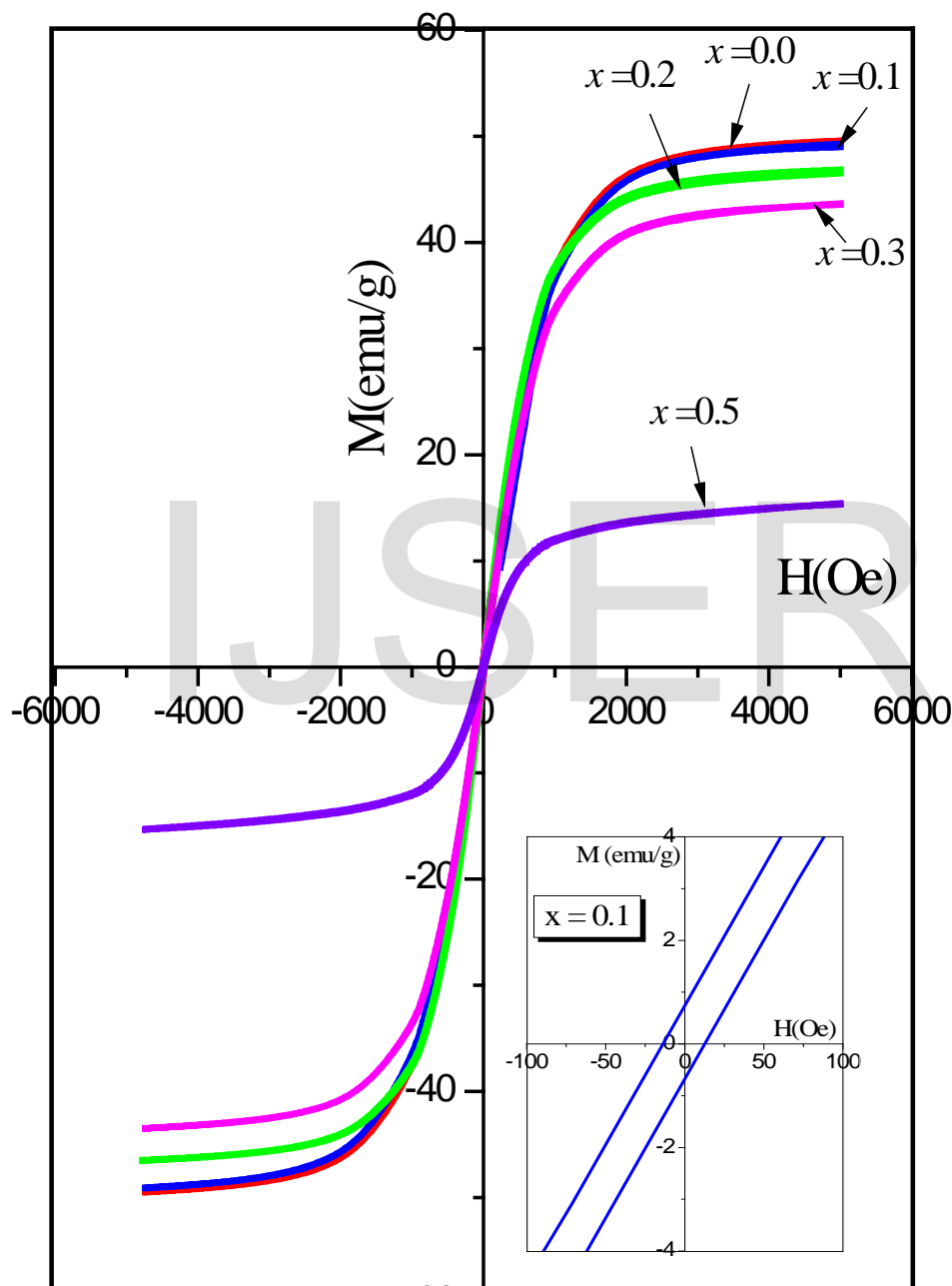


Fig. 7: The DC hysteresis curves for the prepared samples  $Mn_{1-x}Zn_xFe_2O_4$

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### Comparison with the previous work

The obtained results show that, the final sintering of the samples in two steps relatively improves some of the physical properties of



Mn-Zn ferrites compared with the published ones. The in Table [3] and compared with some of the published obtained results for  $\text{Mn}_{0.9}\text{Zn}_{0.1}\text{Fe}_2\text{O}_4$  are summarized data.

**Table 2:** Effect of  $\text{Zn}^{2+}$  ions substitution on the magnetization  $M_s$ , remanence  $M_r$  and coercivity  $H_c$

x	$M_s$ (emu/g)	$M_r$ (emu/g)	$H_c$ (Oe)
0.0	49.2	0.70	13.2
0.1	49.5	0.94	16.6
0.2	46.7	0.63	10.4
0.3	43.5	0.60	11.6
0.5	15.4	0.42	15.6

**Table 3:** The obtained data compared with the published data for  $\text{Mn}_{0.9}\text{Zn}_{0.1}\text{Fe}_2\text{O}_4$ .

work	Crystallite size (nm)	Preparation methods	Saturation magnetization $M_s$ (emu/g)	$H_c$ (Oe)	Curie temperature T(K)
Present work	108	stander ceramic technique	49.5	16.6	537
Hessien et. al [13]	125	solid-state reaction	37.36	6.186	-----
Veena et. al [4]	18.9	chemical co-precipitation techniques	56	14	783
Arulmurugan el. al [10]	11.3	chemical co-precipitation techniques	42.4	12.8	633

## 4. Conclusions

- 1- Sintering conditions of 1300°C for 15h with heating rate 2°C/min is suitable conditions to obtain Mn-Zn ferrites in single phase cubic spinel structure.
- 2- XRD measurements confirm the formation of the prepared samples in single phase cubic spinel structure with crystallite size ranged from 90 to 115 nm.
- 3- Lattice parameter and crystallite size (t) of the prepared samples were decreases with increasing  $\text{Zn}^{2+}$  content.
- 4- EDX measurements show that, the stoichiometric values are very close to the measured values.
- 5- Magnetization was increased up to  $x = 0.1$  and then decreased with increasing  $\text{Zn}^{2+}$  content.
- 6- The Curie temperature of the samples was decreases with increasing  $\text{Zn}^{2+}$  content.

- 7- VSM was investigated at room temperature and shows a hysteresis indicating a ferromagnetic behavior and an improvement in the magnetic properties was observed.
- 8- The obtained results shows that the final sintering of the samples in two steps improves some of the physical properties of Mn-Zn ferrites compared with the published ones.

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